# Thermal characteristics, Raman spectra, optical and structural properties of TiO<sub>2</sub>- $Bi_2O_3$ - $B_2O_3$ -TeO<sub>2</sub> glasses

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# Thermal Characteristics, Raman Spectra, Optical and Structural Properties of TiO<sub>2</sub>-Bi<sub>2</sub>O<sub>3</sub>-B<sub>2</sub>O<sub>3</sub>-TeO<sub>2</sub> Glasses

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Abstract. Tellurite and borotellurite glasses containing Bi<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> were prepared and structure-property correlations were carried out by density measurements, X-ray Diffraction (XRD), Differential Scanning Calorimetry (DSC), Raman and UV-visible spectroscopy. Titanium tellurite glasses require high melt-cooling rates and were fabricated by splat quenching. On adding B<sub>2</sub>O<sub>3</sub>, the glass forming ability (GFA) enhances, and glasses could be synthesized at lower quenching rates. The density of glasses shows a direct correlation with molecular mass of the constituents. UV-visible studies were used to determine the optical band gap and refractive index. Raman studies found that the co-ordination number of tellurium ions with oxygen ( $N_{T_2,O}$ ) decreases with the increase in B<sub>2</sub>O<sub>3</sub> as well as Bi<sub>2</sub>O<sub>3</sub> content while, TiO<sub>2</sub> produce only a small decrease in N<sub>Te-O</sub>, which explains the lower GFA of titanium tellurite glasses that do not contain  $Bi_2O_3$  and  $B_2O_3$ . DSC studies show that the glass transition temperature ( $T_g$ ) increases with  $B_2O_3$  and TiO<sub>2</sub> concentrations and that T<sub>g</sub> correlates well with bond enthalpy of the metal oxides.

Keywords: Tellurite and Borotellurite glasses, X-Ray Diffraction, Differential scanning calorimetry, Raman spectroscopy PACS: 64.70.kj, 61.05.cp, 65.60.+a, 33.20.Fb.

## **INTRODUCTION**

Tellurium oxide-based glasses have found remarkable technological applications due to their unique physical properties such as high refractive index, non-linear optical properties, high dielectric constant, wide optical transmission window [1]. The ab-initio studies show that the higher order of hyperpolarizability is inherent exclusively to the molecules having the form of linear chains formed from 'n' polymerized Te-O species [2].

The ternary TeO<sub>2</sub>-based glassy systems containing TiO<sub>2</sub> modifier, jointly with Bi<sub>2</sub>O<sub>3</sub> produces glasses which offer the best compromise on the high nonlinear optical characteristics and the high mechanical and thermal resistance [3].

Bi<sub>2</sub>O<sub>3</sub> is added to borotellurite glass to increase its density and to improve its structural and optical properties. The borotellurite glasses contain BO<sub>4</sub> tetrahedral, BO3 trigonal units, TeO4 trigonal bipyramidal, TeO3 pyramidal and TeO6 octahedral units at a high  $B_2O_3$  content [4]. The aim of this work is to study the glass forming ability (GFA) of tellurite and borotellurite glasses containing Bi<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> and perform structure-property correlations by Differential Scanning Calorimetry (DSC), UV-Visible and Raman Spectroscopy.

#### **EXPERIMENTAL**

Tellurite and borotellurite glasses containing Bi<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> with composition: xTiO<sub>2</sub>-yBi<sub>2</sub>O<sub>3</sub>-zB<sub>2</sub>O<sub>3</sub>-(100-x-y-z) TeO<sub>2</sub> (x=5, 10-mol%, y=5, 20 and 30-mol %, z=20 and 30-mol %) were prepared using TeO<sub>2</sub> (Aldrich 99%), Bi<sub>2</sub>O<sub>3</sub> (Aldrich 99.9%), H<sub>3</sub>BO<sub>3</sub> (Aldrich 99.9%), TiO<sub>2</sub> (Aldrich 99.9%) by meltquench technique. Appropriate amounts of chemicals were weighed, and put in a platinum crucible after grinding and mixing chemicals. Sintering of the batch mixture was done at 250°C for 24 h in an electric furnace. The temperature of the furnace was then slowly raised to 900°C and the melt was kept at this temperature for 1 h, before quenching it on a heavy brass plate. Glasses without B<sub>2</sub>O<sub>3</sub> were prepared by

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splat quenching as these required higher cooling rates. The samples were characterized by density, XRD, DSC, UV-visible and Raman studies.

### **RESULTS AND DISCUSSIONS**

XRD measurements were performed on Bruker D8 Focus X-ray Diffractometer using Cu  $K_{\alpha}$  radiation in the 2 $\theta$  range of 15° to 70°. Sharp peaks were absent in the XRD patterns but a broad hump in the range of 22° to 34° was present in all glasses, which confirmed the amorphous structure of all samples [Figure 1].

Density measurements were made by Archimedes method on an electronic balance. Density of samples increases from 5.67 g cm<sup>-3</sup> in the sample: 5TiTe to  $6.12 \text{ g cm}^{-3}$  in the sample: 5Ti5BiTe, on incorporating Bi<sub>2</sub>O<sub>3</sub> due to the high molecular weight of Bi<sub>2</sub>O<sub>3</sub> (M.Wt = 465.96 amu) compared to that of TeO<sub>2</sub> (M.Wt = 159.6 amu). It is found that with the increase in the B<sub>2</sub>O<sub>3</sub> concentration (M.Wt = 69.62 amu) and also TiO<sub>2</sub> (M.Wt = 79.90 amu) the density of glasses decreases as these lighter constituents replace the heavier TeO<sub>2</sub>. Variation of density and molar volume along with DSC results are given in Table 1.

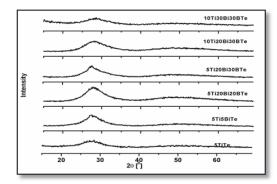
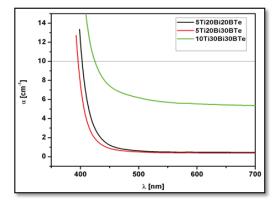


FIGURE 1. XRD patterns of glass samples.

DSC studies for the measurement of thermal properties such as glass transition temperature ( $T_g$ ), crystallization temperature ( $T_c$ ) and the liquidus temperature ( $T_m$ ) was performed on SETARAM SETSYS Evolution-1750 system in the temperature range of 200-750°C at a heating rate of 10°C/min, in

air flow of 20 ml/min.  $T_g$  increases significantly from 323°C to 394°C with increase in  $B_2O_3$  content due to greater bond enthalpy of B-O bonds (804 kJ mol<sup>-1</sup>) compared to that of Te-O bonds (376 kJ mol<sup>-1</sup>). It was found that  $T_g$  increases with increase in the concentration of both TiO<sub>2</sub> and Bi<sub>2</sub>O<sub>3</sub>. The increase in  $T_g$  for both Bi<sub>2</sub>O<sub>3</sub> (337.2 kJ mol<sup>-1</sup>) and TiO<sub>2</sub> (666 kJ mol<sup>-1</sup>) indicates that the Ti-O-Te and Bi-O-Te bonds are stronger than Te-O-Te linkages [3].

UV-visible studies for the measurement of optical absorption spectra of the polished disk shaped glass samples were performed on Shimadzu 1601 double beam UV-visible spectrophotometer in the wavelength range of 200-1100 nm. The absorption coefficient  $\alpha(\lambda)$  was determined by dividing the absorbance A, with the thickness of glass samples and is shown in Figure 2.



**FIGURE 2.** Optical absorption spectra of bismuth and titanium borotellurite glasses.

Optical band gap,  $E_g$  was calculated from the cutoff wavelength  $\lambda_o$ , which was arbitrarily defined as the wavelength at which  $\alpha$ =10 cm<sup>-1</sup>.  $\lambda_o$  shifts to lower wavelengths as the content of B<sub>2</sub>O<sub>3</sub> is increased, while it shifts to larger wavelengths with increase in Bi<sub>2</sub>O<sub>3</sub> mol% [Table 2].

 $E_g$  was used to calculate refractive index,  $\eta$  [3]. With increase in the Bi<sub>2</sub>O<sub>3</sub> content,  $E_g$  decreases and  $\eta$  increases because Bi<sup>3+</sup> ions have high polarity that leads to breaking of bridging oxygen to non-bridging oxygen (NBO). The increase in the band gap with increase in B<sub>2</sub>O<sub>3</sub> content is due to the decrease in the

TABLE 1. Density, DSC and N<sub>Te-O</sub> results of tellurite and borotellurite glasses containing Bi<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub>.

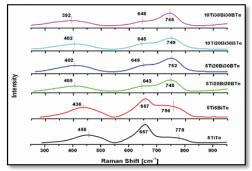
Sample Code	Composition (mol%)				Density(±0.01) (g cm <sup>-3</sup> )	T <sub>g</sub> (°C)	N <sub>Te-O</sub>
	TiO <sub>2</sub>	Bi <sub>2</sub> O <sub>3</sub>	$B_2O_3$	TeO <sub>2</sub>	,		
5TiTe	5	-	-	95	5.67	323	3.65
5Ti5BiTe	5	5	-	90	6.12	328	3.50
5Ti20Bi20BTe	5	20	20	55	6.15	383	3.33
5Ti20Bi30BTe	5	20	30	45	5.93	394	3.31
10Ti20Bi30BTe	10	20	30	40	5.82	406	3.28
10Ti30Bi30BTe	10	30	30	30	6.39	410	3.22

NBO concentration.

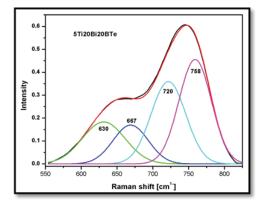
**TABLE 2.** Optical properties of tellurite and borotellurite glasses containing  $Bi_2O_3$  and  $TiO_2$ .

Sample Code	$\lambda_0[nm]$	Eg[eV]	η
5Ti20Bi20BTe	403	3.08	2.37
5Ti20Bi30BTe	396	3.14	2.35
10Ti30Bi30BTe	425	2.92	2.42

Raman studies were performed on samples with Renishaw In-via Reflex micro-Raman spectrometer using 514.5 nm argon ion laser as excitation source, in the wavenumber range: 300 to 1000 cm<sup>-1</sup>. Short-range structural properties of the glasses were elucidated from Raman studies. Raman spectra of glass samples are shown in Figure 3.



**FIGURE 3.** Raman spectra of tellurite and borotellurite glasses containing  $Bi_2O_3$  and  $TiO_2$ .



**FIGURE 4.** Deconvoluted Raman spectra of 5Ti20Bi20BTe glass sample.

Two characteristic Raman bands are observed in the wavenumber range: 400 to 500 cm<sup>-1</sup> and 550 to 830 cm<sup>-1</sup>. With the addition of modifier  $Bi_2O_3$ , the band in the range: 300 to 500 cm<sup>-1</sup> shifts towards 300 cm<sup>-1</sup> thus indicating that the stronger Te-O-Te and Te-O-Ti bridges are replaced by weaker Te-O-Bi linkages, relative to the spectrum of crystalline  $Bi_2Te_4O_{11}$  [4]. The presence of Raman peak at 450 cm<sup>-1</sup> is due to the stretching and bending vibrations of Te-O-Te linkages in TeO<sub>4</sub>, TeO<sub>3+1</sub> and TeO<sub>3</sub> structural units. The band at ~650 cm<sup>-1</sup> is due to the Te-O stretching vibrations in TeO<sub>4</sub>. The peak at 750–760 cm<sup>-1</sup> becomes more prominent with the increasing content of Bi<sub>2</sub>O<sub>3</sub> and is due to the stretching vibrations of Te-O bonds in TeO<sub>3</sub>. This suggests that Bi<sub>2</sub>O<sub>3</sub> causes an increase in TeO<sub>3</sub> at the expense of TeO<sub>4</sub> structural groups.

Raman spectra was de-convoluted and the peaks centered at  $\sim$ 630, 667, 720, 758 cm<sup>-1</sup> were obtained, as shown in Figure 4 for one sample. The area under the peaks was used to calculate Te-O co-ordination using the formula [3]:

$$N_{Te-O} = 3 + \frac{A_{650} + A_{667}}{A_{650} + A_{667} + A_{720} + A_{758}}$$
 1

The Te-O co-ordination number decreases from 3.50 to 3.22 on incorporating  $B_2O_3$ . This decrease in  $N_{Te-O}$  is due to the transformation of  $TeO_4$  into  $TeO_3$  structural units [3,4].

#### CONCLUSIONS

Tellurite and borotellurite glasses containing  $TiO_2$ and  $Bi_2O_3$  were fabricated and characterized by density, XRD, DSC, Raman and UV-visible studies. Density increases with increase in  $Bi_2O_3$  due to its higher molecular weight. Glass transition temperature increases with both  $TiO_2$  and  $Bi_2O_3$ , although Te-O coordination is not significantly modified by  $TiO_2$ . Glass forming ability of tellurite glasses containing  $TiO_2$  and  $Bi_2O_3$  enhances significantly on incorporating  $B_2O_3$ into the tellurite network and is due to the structural transformation:  $TeO_4 \rightarrow TeO_3$ .

### ACKNOWLEDGEMENT

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#### REFERENCES

- 1. R.A. El-Mallawany, *Tellurite Glasses Handbook:* physical properties and data. 2011: CRC Press.
- M. Soulis, T. Merle-Méjean, A.P. Mirgorodsky, O. Masson, E. Orhan, P. Thomas and M.B. Smirnov, *Journal of Non-Crystalline Solids* 354, 199-202 (2008).
- N. Kaur and A. Khanna, *Journal of Non-Crystalline* Solids 404, 116-123 (2014).
- M. Udovic, P. Thomas, A. Mirgorodsky, O. Durand, M. Soulis, O. Masson, T. Merle-Méjean and J.C. Champarnaud-Mesjard, *Journal of Solid State Chemistry* 179, 3252-3259 (2006).